

# Chem 406 - Lecture 7

## Structure Determination by X'Ray Crystallography

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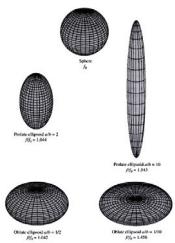
## Introduction

- Earlier in the semester we developed a crude model for macromolecules using sedimentation analysis.
  - Prolate and oblate ellipsoids of revolution

TABLE 5.1 FRICTIONAL COEFFICIENT RATIOS

Shape	$f/f_s$	$R_e$
Prolate ellipsoid	$\frac{P-a}{(P+a)^{1/2}}$	$(ab)^{1/3}$
Oblate ellipsoid	$\frac{(P-a)}{P(a-b)^{1/2}}$	$(a^2b)^{1/3}$
Long rod	$\frac{G}{L+2P} = \frac{G}{2b}$	$(\frac{1}{2}a^2b)^{1/3}$

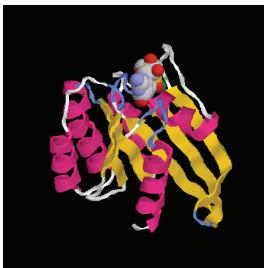
In these equations,  $P = ab$ , where  $a$  is the semimajor axis (or the half-length for a rod) and  $b$  is the minor axis (or radius of a rod).  $R_e$  is the radius of a sphere equal in volume to the ellipsoid or rod, so  $R_e = 4\pi a b^2 / 3V$ .



2

## Introduction

- Since the late 1950's we have had models with atomic resolution.



3

## Protein Data Bank

### HTTP Status 500 -

Exception report  
message  
description: The server encountered an internal error () that prevented it from fulfilling this request.  
exception  

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org.apache.tiles.DefinitionsFactoryException: XML error reading definitions.
	org.apache.tiles.digester.DigesterDefinitionsReader.readDefinitionsReader.java:166)
	org.apache.tiles.digester.DigesterDefinitionsReader.readDefinitionsDefinitionsFactory.java:220)
	com.opensymphony.webwork.views.tiles.TilesResult.getComponentDefinition(TilesResult.java:137)
	com.opensymphony.webwork.views.tiles.TilesResult.doExecute(TilesResult.java:91)
	com.opensymphony.webwork.views.tiles.TilesResult.execute(TilesResultSupport.java:142)
	com.opensymphony.xwork.DefaultActionInvocation.executeResult(DefaultActionInvocation.java:111)
	com.opensymphony.xwork.DefaultActionInvocation.intercept(DefaultActionInvocation.java:164)
	org.rosb.mydb.AuthenticationInterceptor.intercept(AuthenticationInterceptor.java:28)
	com.opensymphony.xwork.DefaultActionInvocation.invoke(DefaultActionInvocation.java:168)
	com.opensymphony.xwork.interceptor.ParametersInterceptor.intercept(ParametersInterceptor.java:131)
	com.opensymphony.xwork.interceptor.TimerInterceptor.intercept(TimerInterceptor.java:118)
	com.opensymphony.xwork.interceptor.StructureAwareInterceptor.intercept(StructureAwareInterceptor.java:61)
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	com.opensymphony.xwork.DefaultActionInvocation.invoke(DefaultActionInvocation.java:168)
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	com.opensymphony.xwork.interceptor.MethodFilterInterceptor.intercept(MethodFilterInterceptor.java:115)
	com.opensymphony.xwork.interceptor.MethodFilterInterceptor.intercept(MethodFilterInterceptor.java:86)
	com.opensymphony.xwork.DefaultActionInvocation.invoke(DefaultActionInvocation.java:168)
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4

Protein Data Bank

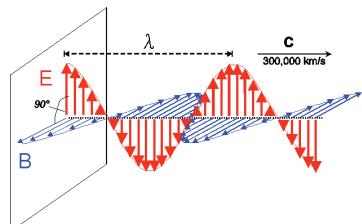
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## Protein Data Bank

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 VERSION 2.10  
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 COMPOUND 2 MOLECULES: SARCOPLASMIC/ENDOPLASTIC RETICULUM CALCIUM  
 COMPOUND 4 CHAIN: A  
 COMPOUND 5 SYNTHON: CALCIUM-TRANSFERRING ATPASE SARCOPLASMIC  
 COMPOUND 6 FUNCTION: TRANSPORT, FAST TWITCH SKELETAL MUSCLE, SERCA1A  
 COMPOUND 7 EC1 4.3.4.8  
 COMPOUND 8 EC2 4.3.4.8  
 SOURCE PDB ID: 1d1  
 SOURCE SCIENTIFIC, SYNTOCYTICUS CUCURBITUS  
 SOURCE 3 ORGANISM: CONUS RABBIT  
 KEYWORD KINETICS  
 KEYWORD ION PUMP, CALCIUM, MEMBRANE PROTEIN, P-TYPP ATPASE, ACTIVE  
 KEYWORD 2 TRANSPORT  
 KEYWORD 3 CALCIUM  
 AUTHOR C.T.YOSHIOKA,M.NAMASAKO,H.NOMURA,H.OGANA  
 JOURNAL JOURNAL OF BIOPHYSICAL STUDIES OF THE CALCIUM PUMP AS A RESOLUTION  
 PAPER 1 14-16 1998  
 PAPER 04-04-1998 14EUL  
 JRNL TITL 2 SARCOPLASMIC RETICULUM AT P.D. V. 405 647 2000  
 JRNL TITL 2 SARCOPLASMIC RETICULUM AT P.D. V. 405 647 2000  
 REFN 1 REFN 1 ATOM NATURE US ISSN 0282-0814  
 REMARK 1 REFERENCE 1  
 REMARK 1 AUTH C.T.YOSHIOKA,H.NOMURA  
 REMARK 1 TITL STRUCTURAL CHANGES IN THE CALCIUM PUMP  
 REMARK 1 FUNCTION: TRANSFERRING THE DISOCIATION OF CALCIUM.  
 REMARK 1 REFN 1  
 REMARK 1 AUTH C.T.YOSHIOKA,H.NOMURA,Y.SUGI  
 REMARK 1 TITL 2 SARCOPLASMIC RETICULUM AT P.D. V. 418 605 2002  
 REMARK 1 REFN 1  
 REMARK 1 AUTH C.T.YOSHIOKA,H.NOMURA,Y.SUGI  
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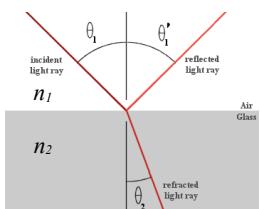
6

## Electromagnetic Radiation



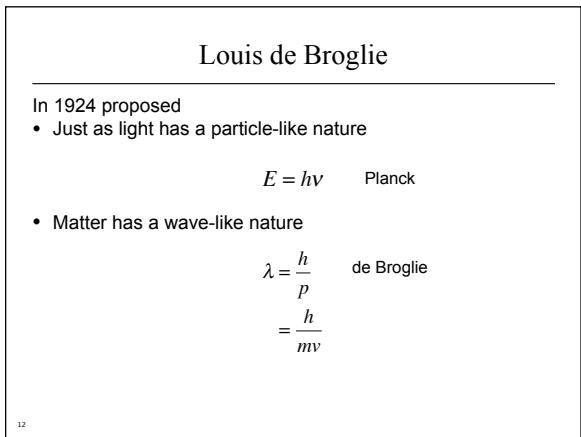
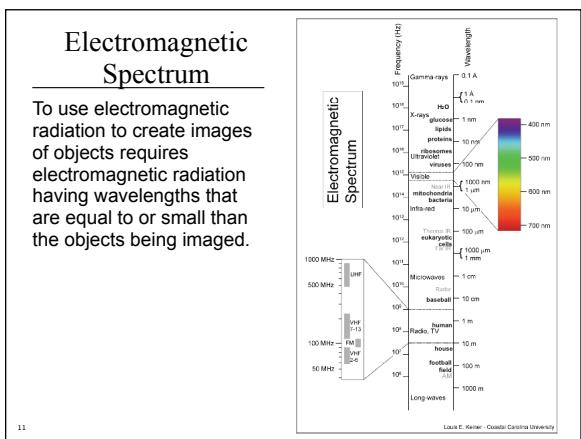
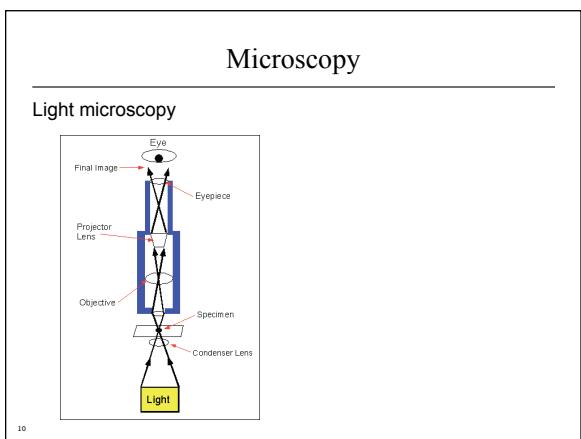
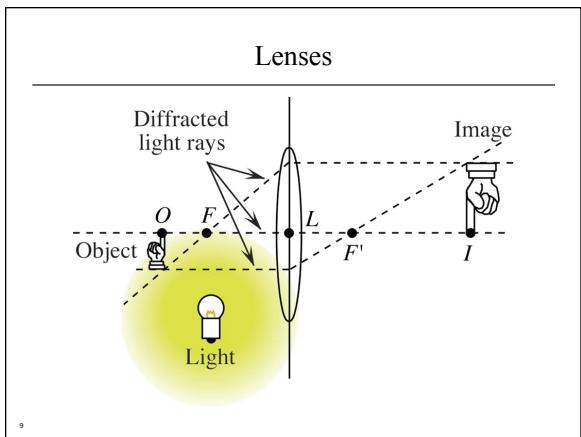
7

## Snell's Law



$$n_1 \sin(\theta_1) = n_2 \sin(\theta_2)$$

8



## High Energy Electron Beam

- For a 100 kV beam, in which an electron is accelerated through a 100,000 V potential field.

$$\begin{aligned}\lambda &= \frac{h}{mv} \\ &= 0.004 \text{ nm} \\ &= 0.04 \text{ \AA}\end{aligned}$$

13

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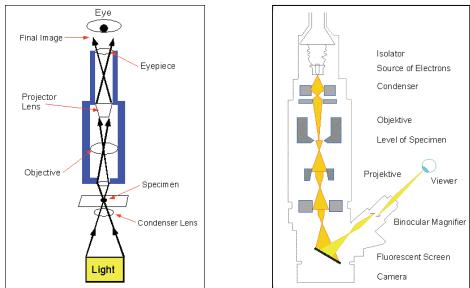
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## Microscopy

### Comparing light microscopy to electron microscopy



14

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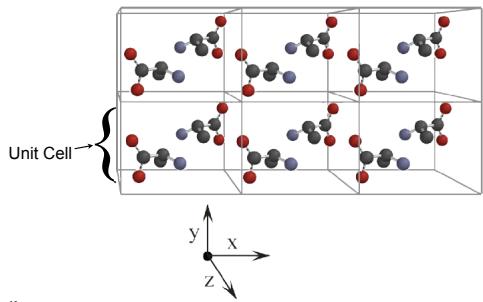
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## X-ray Crystallography

### The Crystal Lattice



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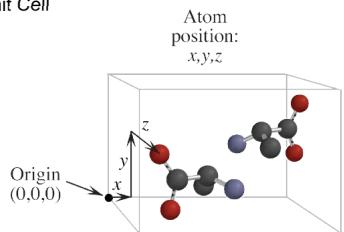
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## X-ray Crystallography

### The Unit Cell



16

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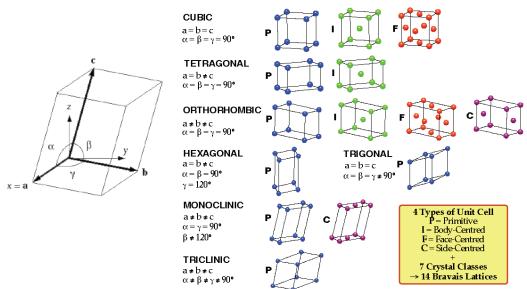
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## X-ray Crystallography

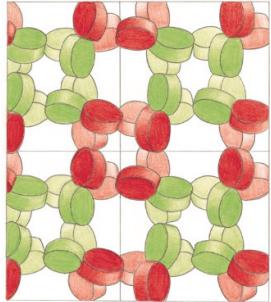
### The Bravais Lattice Types



17

## X-ray Crystallography

### Protein Crystals



18

## X-ray Crystallography

### Protein Crystals

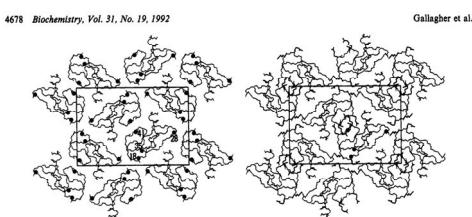
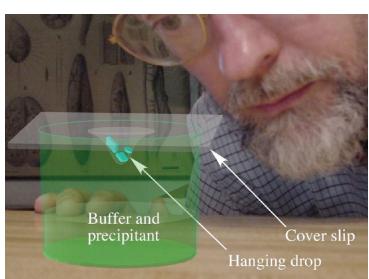


FIGURE 4. Packing of BPTI molecules in crystal form III. (Left figure) Backbone C, CA, and N atoms are shown in projection down the crystallographic *c* axis. Closed circles indicate the NH of Ile 18, Tyr 35, Gly 28, and Lys 41. (Right figure) Side chains at the two major contact sites are shown in bold. The residues involved in the salt bridge of Arg 39, Lys 41, and Arg 42 are in the center and at the corners of the backbone projection. Asp 17 and Glu 26 are at the edges of the sheet. Closed circles mark the location of 18, 35, 38, and 41 NH in one molecule. BPTI form III structure is reported in Wlodawer et al. (1987b).

19

## X-ray Crystallography

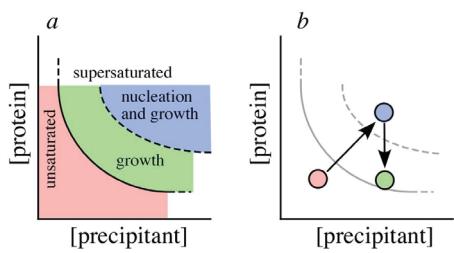
### Growing Crystals



20

## X-ray Crystallography

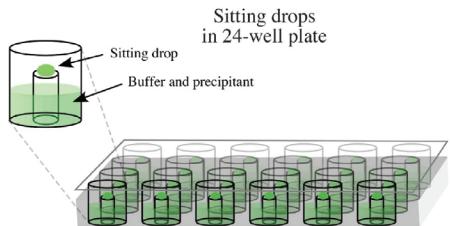
### Growing Crystals



21

## X-ray Crystallography

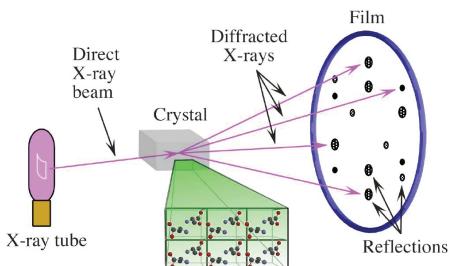
### Growing Crystals



22

## X-ray Crystallography

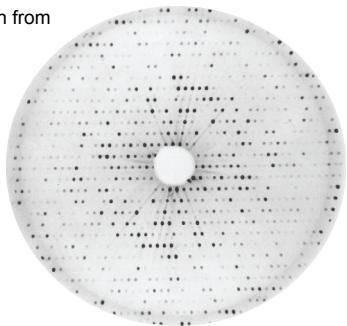
### Diffraction from Crystals



23

## X-ray Crystallography

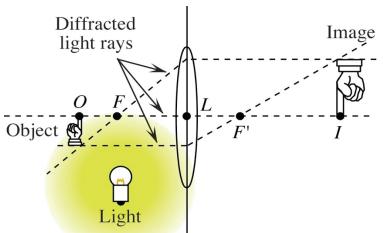
### Diffraction from Crystals



24

## X-ray Crystallography

In light microscopy, lenses are used to collect the light scattered from an object and use it to form an image.

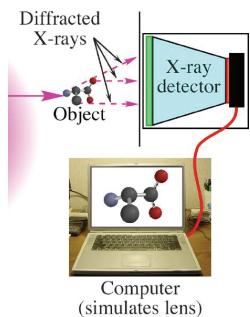


25

## X-ray Crystallography

There are no X-ray lenses

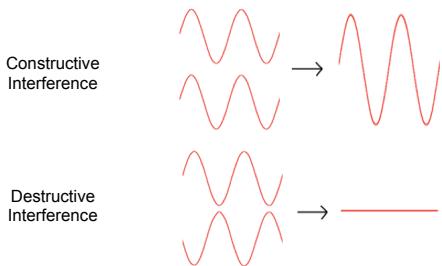
- Computers are used to simulate a lens
- To understand how this works, we need to understand the elements of a diffraction pattern



26

## The Diffraction Pattern

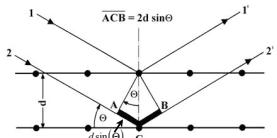
Constructive and Destructive Interference



27

## The Diffraction Pattern

Bragg's Law describes the conditions required for obtaining a spot.



- Constructive interference occurs when  
$$2d \sin(\Theta) = n\lambda \quad (\text{Bragg's Law})$$

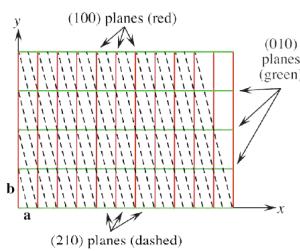
- William Henry Bragg, and his son William Lawrence Bragg, shared the [Nobel Prize in Physics in 1915](#).

28

## The Diffraction Pattern

### Crystal Diffraction

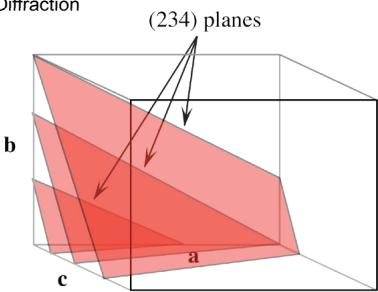
- The spots arise from reflections off of the crystal planes



29

## The Diffraction Pattern

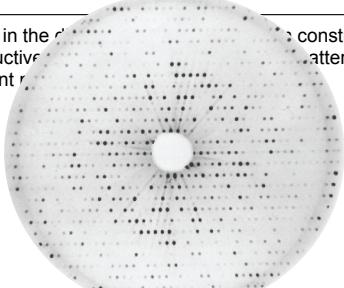
### Crystal Diffraction



30

## The Diffraction Pattern

### The spots in the diffraction pattern arise from constructive interference of waves scattered from the different reflection planes.



constructive interference of waves scattered from the different reflection planes.

- The spacings and intensities of the spots give the unit cell dimensions and angles ( $a$ ,  $b$ ,  $c$ ,  $\alpha$ ,  $\beta$  &  $\gamma$ )

31

## The Diffraction Pattern

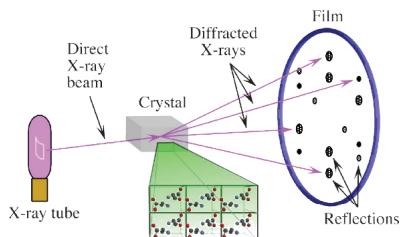
### Demos

- XRayView 3.0
  - Simulates diffraction from a crystal
- Optical Diffraction
  - Diffraction patterns are reciprocally related to the object responsible for the diffraction

32

## The Diffraction Pattern

The diffraction pattern is also referred to at the *reciprocal lattice*.



33

## The Fourier Series

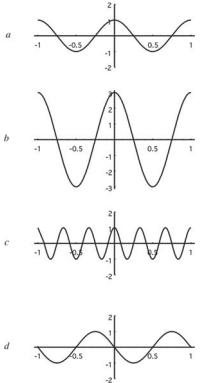
### Fourier Series

- Sine and Cosine waves have
  - Amplitude,  $F$
  - Frequency,  $h$
  - Phase,  $\alpha$

$$f(x) = F_o \cos(2\pi[hx + \alpha])$$

or

$$f(x) = F_o \sin(2\pi[hx + \alpha])$$



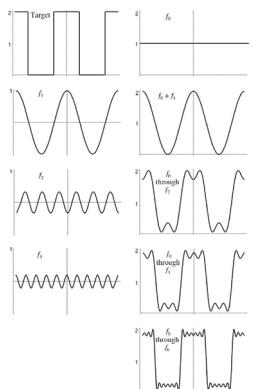
34

## The Fourier Series

### Fourier Synthesis

- Approximating a square wave by a Fourier Series

$$\begin{aligned} f(x) &= F_0 \cos(2\pi[0x + \alpha_0]) \\ &+ F_1 \cos(2\pi[1x + \alpha_1]) \\ &+ F_2 \cos(2\pi[2x + \alpha_2]) \\ &+ F_3 \cos(2\pi[3x + \alpha_3]) \\ &\dots \\ &+ F_n \cos(2\pi[nx + \alpha_n]) \\ &= \sum_{h=0}^n F_h \cos(2\pi[hx + \alpha_h]) \end{aligned}$$



35

## Fourier Series

An alternative to designating a phase angle,  $\alpha$ , is to combine cosine and sine wave

- The sine is equal to a cosine which has been shifted by an angle of  $\pi/2$  ( $90^\circ$ )

$$\sin(x) = \cos(x + \pi/2)$$

- This allows us to replace

$$f(x) = \sum_{h=0}^n F_h \cos(2\pi[hx + \alpha_h])$$

with

$$f(x) = \sum_{h=0}^n (F_{h,\cos} \cos(2\pi[hx]) + F_{h,\sin} \sin(2\pi[hx]))$$

36

## Fourier Series

It is also mathematically convenient to represent this relationship using complex numbers.

$$x = a + ib, \text{ where } i = \sqrt{-1}$$

$$f(x) = \sum_{h=0}^n F_h (\cos(2\pi hx) + i \sin(2\pi hx))$$

where the amplitude,  $F_h$ , is also a complex number

- The cosine portion represents the *Real* part, while the sine portion represents the *Imaginary* part, of the complex number

37

## Fourier Series

Another mathematical convenience comes from making use of Euler's formula:

$$\cos(x) + i \sin(x) = e^{ix}$$

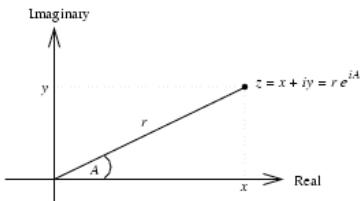
- This allows us to cast the Fourier series into a nice, compact, complex exponential form:

$$f(x) = \sum_{h=0}^n F_h e^{i2\pi hx}$$

38

## Fourier Series

The complex polar coordinate system



39

## Applying the Fourier Series to Diffraction

- The diffraction pattern also called the *reciprocal lattice*.
- Each spot in the diffraction pattern is called a structure factor

- The structure factors are represented in three dimensions by the symbol

$$F_{h,k,l}$$

- Where  $h$ ,  $k$ , and  $l$  are the *Miller indices*, which describe the sets of planes that can be used to divide up the unit cell in the crystal lattice.

40

## Applying the Fourier Series to Diffraction

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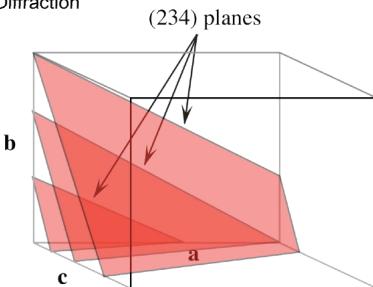
- Where  $h$ ,  $k$ , and  $l$  are the *Miller indices*, which describe the sets of planes that can be used to divide up the unit cell in the crystal lattice.

► Remember, According to Bragg's Law, it is the reflections from these planes that give rise to the diffraction pattern.

41

## Applying the Fourier Series to Diffraction

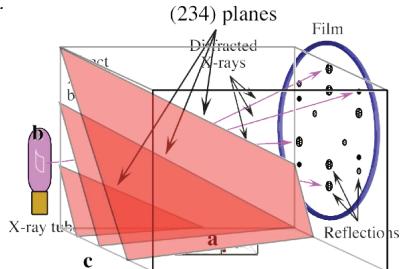
### Crystal Diffraction



42

## Applying the Fourier Series to Diffraction

The diffraction pattern is also referred to at the *reciprocal lattice*.



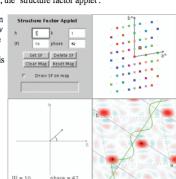
43

## The Interactive Structure Factor Tutorial

Welcome to the structure factor tutorial! The aim of this tutorial is to learn about structure factors, phases, symmetry, and the relationship between the structure factor and the electron density map. The material uses an interactive tool, the "structure factor applet".

The structure factor applet has four windows. Top left is the control window, in which you can select reflections and set their values by entering values in the text boxes. Several buttons allow you to modify the structure factor values in the map and the diffraction pattern. Top right is the structure factor window, which shows the current diffraction pattern. Bottom left is the reflection window, with which you can view or modify magnitudes and phases. Bottom right is the map window, showing the unit cell and electron density.

Magnitudes and phases are encoded as colours using the same scheme as the [Book of Fourier](#).



**Small screen  
or  
slow browser**

There are two versions of the structure factor tutorial. If you have a small screen (less than 1280x1024), or if you have a browser with a slow Java virtual machine (e.g. SGI machines), pick the links on the left.

**Millies.** [Structure factor\(1\)](#), [Structure factor\(2\)](#), [Symmetry\(1\)](#), [Symmetry\(2\)](#), [Patterson\(1\)](#), [Patterson\(2\)](#), [Direct Methods](#).  
**Big screen  
and  
fast browser**

If you have a big screen (at least 1280x1024), and a browser with a fast Java virtual machine (e.g. Sun, Linux, Windows), pick the links on the right.

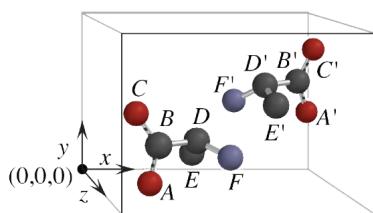
If you are not sure about your machine, try both versions and use whichever works better for you.

**Home and Back**

44

## Applying the Fourier Series to Diffraction

The structure factors (spots),  $F_{h,k,l}$

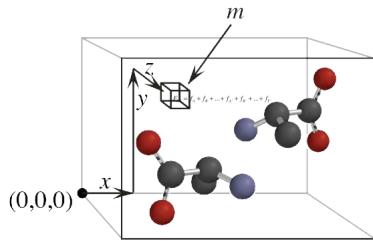


$$F_{hkl} = f_A + f_B + \dots + f_{A'} + f_{B'} + \dots + f_{F'} \quad \text{Sum over atoms}$$

45

## Applying the Fourier Series to Diffraction

The structure factors (spots),  $F_{h,k,l}$



$$F_{hkl} = f(\rho_1) + f(\rho_2) + \dots + f(\rho_m) + \dots + f(\rho_n) \quad \text{Sum over volume elements}$$

46

## Applying the Fourier Series to Diffraction

The structure factor can be related to the electron density using a complex exponential Fourier Series

- Each spot in the diffraction pattern is associated with a particular structure factor,  $F(hkl)$

$$F(hkl) = \sum_x \sum_y \sum_z \rho(xyz) e^{2\pi i(hx+ky+lz)}$$

- Every location in the unit cell contributes to each structure factor
  - If the electron density fluctuates with position at a frequency corresponding to  $h$ ,  $k$  and  $l$ , then this will produce a strong intensity for this particular structure factor,  $F(hkl)$

47

## Fourier Transforms

When one function represents the amplitude for the Fourier series of a second function then the dependent variables for the two functions are reciprocally related.

- In X-ray diffraction, the dependent variables for the electron density function,  $x$ ,  $y$  and  $z$ , have dimensions of distance, whereas the dependent variables for the structure factor,  $h$ ,  $k$  and  $l$ , have dimensions of frequency, or one over distance.
- When this situation exists, there exists a mathematical operation called the *Fourier Transform*, can be used to convert between the two functions

48

## Fourier Transforms

For example, for a one dimensional, discrete function,  $F(h)$ , which is represented as the Fourier series of a second function,  $f(x)$ :

$$F(h) = \sum_{x=0}^{\infty} f(x) e^{i2\pi hx}$$

- The *Fourier transform* is given as

$$f(x) = \sum_{h=0}^{\infty} F(h) e^{-i2\pi hx}$$

49

## Fourier Transforms

When this is extended to 3-dimensions:

$$F(hkl) = \sum_x \sum_y \sum_z f(xyz) e^{i2\pi(hx+ky+lz)}$$

- The *Fourier transform* is given as

$$f(xyz) = \sum_h \sum_k \sum_l F(hkl) e^{-i2\pi(hx+ky+lz)}$$

50

## Fourier Transforms

Substituting  $\rho(xyz)$  for  $f(xyz)$  we get:

$$F(hkl) = \sum_x \sum_y \sum_z \rho(xyz) e^{i2\pi(hx+ky+lz)}$$

- The *Fourier transform* is given as

$$\rho(xyz) = \sum_h \sum_k \sum_l F(hkl) e^{-i2\pi(hx+ky+lz)}$$

- This provides us with a method for determining the electron density function,  $\rho(xyz)$ , from the structure factors,  $F(hkl)$

51

## Solving for the Electron Density Functions

$$\rho(xyz) = \sum_h \sum_k \sum_l F(hkl) e^{-i2\pi(hx+ky+lz)}$$

As indicated earlier, each structure factor  $F(hkl)$  is itself a complex number that is given by an amplitude, phase and frequency

- The Interactive Structure Factor tutorial demonstrates this.

52

## Solving for the Electron Density Functions

- The frequency is determined by the Miller indices  $(h,k,l)$ , which determine the frequency of the planes cutting through the unit cell.
- The amplitude of each structure factor  $|F(hkl)|$ , can be determined from the intensity of each spot  $I(hkl)$ :  
$$|F(hkl)| = \sqrt{I(hkl)}$$
- What's missing is the phase information
  - We have essentially taken a black-and-white photo instead of a colored one

53

## Solving the Phase Problem

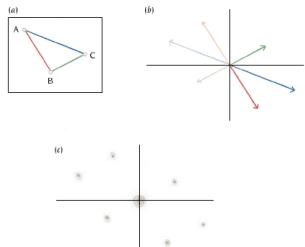
For unit cells with small numbers of atoms, a Patterson Map can be used to determine the distances and directions between the atoms in the unit cell.

- A Patterson Map is constructed by assuming the phases are all zero  
$$P(xyz) = \sum_h \sum_k \sum_l |F(hkl)| e^{-i2\pi(hx+ky+lz)}$$
- [The Interactive Structure Factor tutorial](#) demonstrates this.

54

## Solving the Phase Problem

- Patterson Map



55

## Solving the Phase Problem

For protein crystals, can approximate this situation by using isomorphic replacement

- A small number of metal ions are introduced into the crystal.

Another method used to solve for the structure factor phases is *Molecular Replacement*

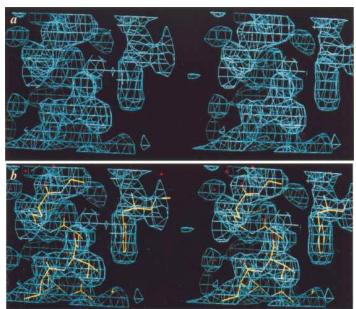
- In this method a homologous protein with a known structure is packed into the cell and used to determine the phases using an *Inverse Fourier Transform*

$$F(hkl)_{\text{calc}} = \sum_x \sum_y \sum_z \rho(xyz) e^{i2\pi(hx+ky+lz)}$$

56

## Model Building

A model is fit to the electron density:



57

## Model Building

The model can be used to calculate a new set of phases:

$$F(hkl)_{calc} = \sum_x \sum_y \sum_z \rho(xyz)_{model} e^{i2\pi(hx+ky+lz)}$$

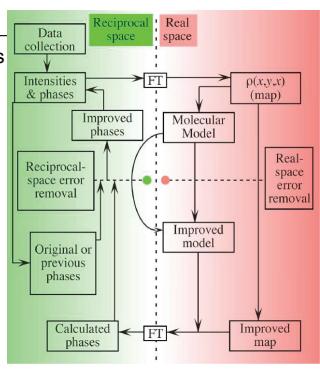
- And this is used to determine a calculated structure factor that can be compared to the observed structure factor:

$$R = \frac{\sum |F(hkl)_{obs}| - |F(hkl)_{calc}|}{\sum |F(hkl)_{obs}|}$$

58

## Model Building

This is an iterative process



59